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## Development of fibrous PLGA/fibrin scaffolds as a potential skin substitute

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The aim of this study has been to fabricate a hybrid electrospun nanofibrous scaffold composed of PLGA/fibrin polymers to be used as a skin substitute and analyze its physical and biological properties. Fibrin (Fib) was obtained from rat blood plasma, characterized and solubilized in formic acid. The final electrospinning solution concentration was 40 % PLGA (w/v) and 1 % fibrin (w/v). To improve spinnability, 3 % PEG (w/v) was added. The scaffolds were characterized by scanning electron microscopy (SEM) and Fourier-transform infrared spectroscopy (FTIR). Water contact angle, maximum elongation, thermal stability, degree of swelling, blood compatibility, cytotoxicity and cell viability were analyzed. The characterization by SEM showed randomly oriented nanofibers with a mean diameter of  $639.8 \pm 241.8$  nm for the PLGA/Fib and  $1051.0 \pm 290.2$  nm for the PLGA. FTIR spectra confirmed the presence of fibrin in the mats. Fibrin incorporation reduced the water contact angle from  $118.9 \pm 2.9$  to  $111.1 \pm 2.8$ . The fibrin increased tensile strength and decreased elongation at break. The scaffolds demonstrated blood compatibility and fibrin incorporation improved cell adhesion and viability when direct and indirect MTT analyses were carried out. Thus, it can be concluded that the PLGA/fibrin mat is a promising material for use as a skin substitute.

**Keywords:** electrospinning; poly(lactic-co-glycolic) acid; fibrin.

## 1. Introduction

Skin is the largest organ of the body, with a surface of 2 m<sup>2</sup>, this being a physical barrier between external and internal environments. It offers protection from dehydration, external damage, infections, UV radiation and environmental toxins. Furthermore, skin acts as a sensory organ, which transmits sensation to pain, temperature and deep pressure, playing an important role in the balance of body temperature and water content and, consequently, in the maintenance of physiological homeostasis (1). Skin is also one of the most immune active organs and interferes in vitamin D production, essential for calcium absorption and normal bone metabolism (2).

Skin disease prevalence has rapidly increased since 2005, with chronic wounds being a growing problem worldwide (3). Due to the importance of preserving skin integrity, tissue engineering has become an alternative to the conventional treatments for full-thickness wounds, which consist of the use of autografts, allografts or xenografts. These conventional treatments have some disadvantages, such as the risk of disease transmission, high cost and donor site morbidity (4).

One of the elements of tissue engineering which are of important relevance are the scaffolds, consisting of biomaterials fabricated from polymers of natural or synthetic origin which supports the cells. Their importance lies in the fact that the composition and structure of such biomaterials can provide appropriate conditions for the tissue regeneration process. Therefore, scaffolds act as an extracellular matrix analog; capable of guiding cell adhesion, proliferation and differentiation and also the promotion of angiogenesis. Such cell functions are influenced by polymer composition and related mechanical properties, including thickness and pore size (5).

Although tissue engineering of the skin has been widely studied, an ideal skin substitute has still not been developed. For a skin substitute to be considered ideal, it must have characteristics

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3 such as water loss prevention because a moist wound environment is required to facilitate the  
4 healing process (6). However, adequate moisture balance is necessary. The water absorption  
5 capacity of the scaffolds, therefore, should be in agreement with the quantity of the wound  
6 exudate, it being possible to maintain a healthy physiological level of moisture (7). Thereby, dry  
7 wounds will require a hydrating scaffold, and in contrast, wounds with excessive exudate will  
8 require an absorptive scaffold able to prevent the maceration of the surrounding skin.  
9 Absorptive scaffolds should be able to transmit moisture vapor away from the wound bed  
10 towards the upper surface and allow the moisture to evaporate into the air. The transmission of  
11 moisture vapor is directly proportional to the porosity of the scaffold (8). Therefore, the scaffold  
12 should prevent dehydration and also the accumulation of extra exudates, such as seroma. For  
13 these reasons, the election of the scaffolds should be in agreement with the characteristics of the  
14 wound. On the other hand, complications like hematomas should be avoided by having good  
15 homeostasis before applying the substitute.

16  
17 Besides water loss balance, characteristics such as a wide availability of the polymer,  
18 biodegradability, support for cellular adhesion, infection resistance, presence of epidermal and  
19 dermal components, low antigenicity, high shear strength, easy storage, long shelf life and good  
20 cost-benefit are desirable (5). These characteristics are dependent on the polymer properties and  
21 the technique used for scaffold fabrication. In this study, electrospinning was applied, which is a  
22 versatile technique widely used for fabricating fibers ranging from the nano- to the microscale.

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24 In addition, the use of hybrid materials has been suggested in order to take advantage of the  
25 physical properties of synthetic polymers as well as the biological ones of natural polymers (9).  
26 Fibrin is a natural polymer composed of fibrinogen monomers present in blood plasma, with  
27 interesting properties to be used in tissue engineering which play an important role in the wound  
28 healing process. Fibrin acts as a support for the hemostatic plug and it is also considered to be a  
29 preliminary matrix within the wound space (10). Fibrin can interact with cells such as  
30 leukocytes, platelets, endothelial cells and fibroblast via integrin receptors. Arg-Gly-Asp (RGD)  
31 motif is the principal integrin-binding domain of ECM proteins, which play an important role in  
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3 cell adhesion (11). It has been shown that the fibrinogen molecules themselves (constituent A  
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5 alpha and B beta chains) promote fibroblast proliferation (12).  
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8 Fibrin not only interacts with cells but also contains motifs that allow for the interaction with  
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10 other ECM proteins, providing an adequate environment for cell adhesion, proliferation and  
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12 migration (13). Native collagen I has binding sites that are common for fibrin, fibronectin and  
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14 metalloproteinase 1 (14). This fact presupposes that a scaffold containing fibrin could provide a  
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16 functional articulation between itself and the native tissue, therefore facilitating cell migration.  
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18 Moreover, there are other ECM proteins capable of binding fibrin and regulating cell adhesion,  
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20 proliferation and migration, such as fibronectin, vitronectin and thrombospondin (13) (11).  
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24 Furthermore, fibrin is an important signaling molecule, being a potent vascular endothelial  
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26 cell activator, inducing the expression of the IL8, which influence leukocyte migration and  
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28 activation (11). It has also been reported that fibrin degradation product D-dimer induces  
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30 synthesis and the release of IL-1 $\beta$ , IL-6 and plasminogen activator inhibitors from monocytes *in*  
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32 *vitro*, which are regulators of hepatic response. IL-6 stimulates transcription of acute-phase  
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34 proteins genes and fibrinogen synthesis, meanwhile, IL-1 down-regulates fibrinogen synthesis;  
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36 which means that fibrin plays an active role in the tissue regeneration/repair processes (15).  
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40 Additionally, poly(lactic-co-glycolic) acid (PLGA) is a synthetic polymer approved by the  
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42 European Medicine Agency as well as the US Food and Drugs Administration. PLGA is  
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44 frequently used worldwide for biomedical applications and drug delivery as it is biodegradable,  
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46 biocompatible and its degradation products are easily metabolized through the Krebs Cycle  
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48 (16).  
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51 The aim of this study has been to produce electrospun nanofibrous scaffolds composed of  
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53 PLGA/fibrin polymers and to analyze their physical and biological properties.  
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## 58 **2. Materials and methods**

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## 2.1 Materials

The products used from Sigma-Aldrich were 1,1,1,3,3,3-hexafluoro-2-propanol (HFIP), 3-(4, 5-dimethylthiazol-2-yl)-2,5-diphenyl tetrazolium bromide (MTT), 4,6-diamidino-2-phenylindole (DAPI), Dulbecco's Modified Eagle's Medium (DMEM/HEPES) - low glucose and high glucose, paraformaldehyde (PFA), rhodamine-phalloidin, trypsin-EDTA solution 10x and Triton X-100. Formic acid was purchased from Labsynth. Collagenase and penicillin/streptomycin were purchased from Gibco, PLGA 75:25 from PURAC, Fetal Bovine Serum (FBS) heat-inactivated from Cultilab, Campinas/SP and dimethyl sulfoxide (DMSO) from Nuclear.

## 2.2 Fibrin obtention and characterization

7 male Wistar rats aged 4 months (300–450 g body weight) were obtained from the Animal House of the Institute of Basic Health Sciences of the Federal University of Rio Grande do Sul. All the procedures were performed following authorization from the Ethics Committee (Number 36484, approved on 16th May, 2019).

The blood was collected after anesthesia with citrate-phosphate-dextrose anticoagulant in a 9:1 proportion by cardiac puncture. The harvested blood was centrifuged (800 g, 13 min at 4 °C) and the blood plasma was removed by aspiration, Fig 1. Fibrinogen polymerization was induced by adding 2.5 %  $\text{CaCl}_2$  to the plasma (1:9). The plasma was incubated until clot formation; the clot was placed between two 0.2  $\mu\text{m}$  pore size membranes and weight was applied until the formation of a thin sheet of fibrin. The fibrin was dehydrated at 37 °C for 48 hours. The dehydrated product was analyzed by Fourier transform infrared spectroscopy.

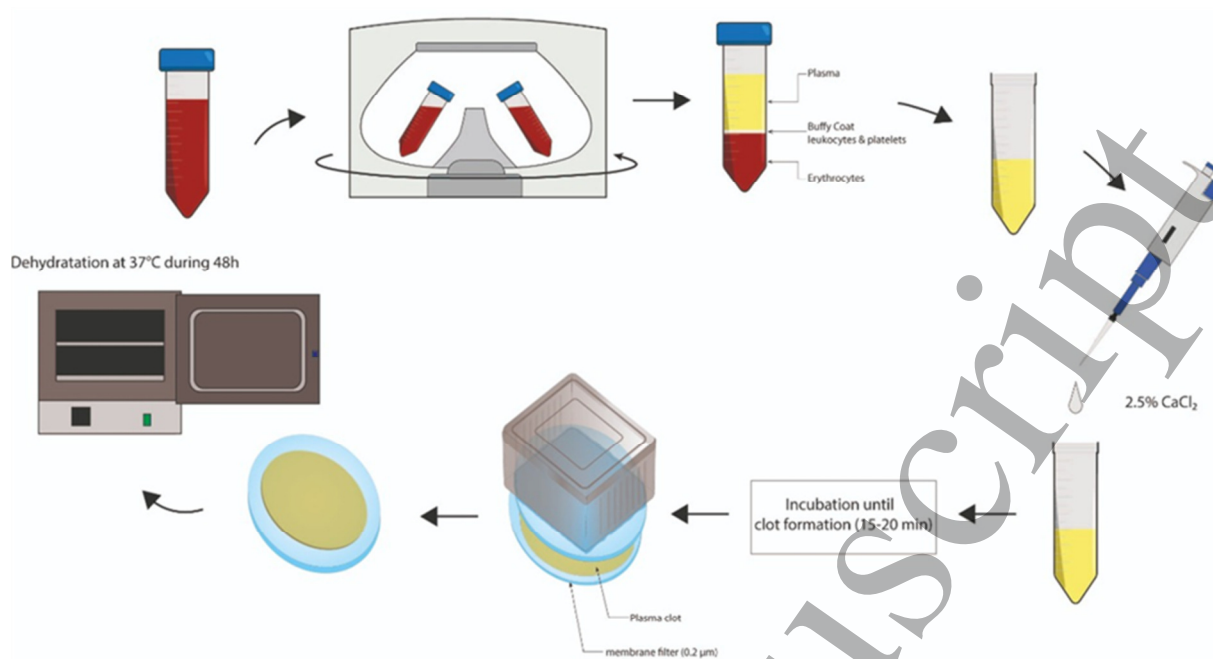


Fig. 1. Schematic process for fibrin obtention.

### 2.3 Preparation of electrospinning solution

Fibrin was dissolved in formic acid at room temperature with continuous stirring overnight at 1000 rpm. Once fibrin was dissolved, HFIP was added at the proportion of formic acid:HFIP 1:1. The solution was stirred for 15 minutes before adding the PLGA in a final solution concentration of 40 % PLGA (w/v), 1 % fibrin (w/v) and 3 % PEG (w/v). To dissolve the PLGA, the solution was stirred for 2 hours at room temperature until homogenization was reached. The electrospinning solution for the control scaffolds was prepared with the same solvents and solute concentration, with the exception of the fibrin.

### 2.4 Fabrication of PLGA and PLGA/fibrin electrospun nanofibers

The polymeric solution was loaded in a 2 ml plastic syringe fitted with a 23 G needle. The electrospinning parameters were fixed as applied high-voltage of 17 kV (18 kV at the positive and 1 kV at the negative) the flow rate of 0.48 ml/h controlled by a syringe pump (electrospinning model homemade) and a tip-collector distance of 20 cm. Random fibers were

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3 electrospun on round glass coverslips of 13 mm diameter and were fixed using the same  
4 polymer solution prepared for the nanofibers fabrication on the underside of the glass coverslips  
5 so that they could be manipulated without distorting the microstructure. The scaffolds were  
6 sterilized with UV light for 1 hour at each site in a laminar flow hood.  
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## 11 **2.5 Characterization**

### 12 **2.5.1 Scanning electron microscopy**

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15 The scaffold morphology and PLGA/Fib fiber diameter, together with the cell presence, were  
16 performed by the SEM model Pro X (Phenom-World – Thermo Fisher Scientific, the  
17 Netherlands). The samples were washed with phosphate buffer (PBS, pH 7.4), fixed in 4 % PFA  
18 for 30 minutes, rinsed in buffer PBS and dehydrated in ethanol series. The dried samples were  
19 metalized with a thin layer of gold and placed on stubs using adhesive carbon disks. Images  
20 were acquired using an accelerating voltage of 15 kV and a magnification range of 500–5,000×.  
21 The average fiber diameter was determined by SEM through the measurement of a total of 200  
22 fibers of PLGA and PLGA/Fib, selected randomly from 7 images for each, using the ImageJ  
23 software.  
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### 41 **2.5.2 Thermal stability and water contact angle**

42 Thermal stability of the scaffolds and residual solvent content was evaluated by a  
43 thermogravimetric analyzer (TGA Q50, TA Instruments, USA), at a heating rate of 20 °C/min,  
44 in a temperature range of 30-800 °C under a nitrogen atmosphere. To analyze hydrophilicity,  
45 the water drop contact angle was carried out by Phoenix Mini (Surface & Electro-Optics Co.,  
46 South Korea) using a sessile drop method (between 0.03 ml and 0.05 ml of drop volume), and a  
47 minimum of 15 drop images were recorded for each sample.  
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### 58 **2.5.3 Mechanical testing**



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3 Maximum elongation and stress were determined by dynamic mechanical analysis (DMA) in a  
4 DMA 2980 (TA Instruments, USA) using the tension film clam. The scaffolds were cut into  
5 rectangular shapes (10×6 mm). The assays were carried out at a constant temperature (37 °C)  
6 with a ramp force of 0.1 N/min until 13 N maximum loads. Data from the stress-strain curves  
7 were recorded and the tensile stress at maximal load was obtained from this data for each  
8 sample. The analyses were made using the TA Universal Analysis software.  
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#### 19 **2.5.4 Fourier transform infrared spectroscopy**

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21 Influences of fibrin in the PLGA scaffolds were analyzed in the fingerprint region in the range  
22 of 2000 and 750 cm<sup>-1</sup> in a FTIR spectrometer Spectrum 1000 (PerkinElmer, Inc., USA) by  
23 averaging 32 scans at a resolution of 4 cm<sup>-1</sup>. The horizontal ATR (HATR) accessory (Pike  
24 Technologies, USA) was used for the PGLA and PLGA/Fib films; the fibrin powder was mixed  
25 with potassium bromide (KBr) (Sigma-Aldrich, USA) and pressed into a disk shape.  
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#### 35 **2.5.5 Swelling test**

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37 The scaffolds were cut into quadrangular shapes (1 cm x 1 cm) and weighed ( $W_i$ ). To calculate  
38 the swelling degree, the membranes were immersed in PBS at 37 °C. After 1 hour and 24  
39 hours, the scaffolds were removed and the surface water blotted out by filter paper to be  
40 weighed after swelling ( $W$ ) (17). The swelling degree was calculated as follows:  
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$$48 \text{Degree of Swelling (\%)} = \frac{W - W_i}{W_i} \times 100$$

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#### 52 **2.5.6 Blood compatibility test**

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54 To test the compatibility of the scaffolds with blood, the hemolysis assay was performed.  
55 Citrated rat blood (4 ml) was diluted with 5 ml of normal saline 0.9 %. For the positive control,  
56 10 ml of Milli-Q water was used and 10 ml of normal saline 0.9 % was taken as the negative  
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control. Scaffolds of diameter 1 cm were submerged in 10 ml of normal saline. Following this, 0.2 ml of citrated diluted blood was added to each group and incubated at 37 °C for 60 minutes, followed by centrifugation at 1500 rpm for 10 minutes. Subsequently, the absorbance of the supernatant was taken at 570 nm. The hemolysis percentage was calculated as follows:

$$\% \text{ hemolysis} = \frac{A_S - A_{NC}}{A_{PC} - A_{NC}}$$

Where  $A_S$  is the absorbance of the sample,  $A_{NC}$  is the absorbance of the negative control,  $A_{PC}$  is the absorbance of the positive control.

### 2.5.7 *In vitro* cell culture studies

Stem cells were isolated from exfoliated human deciduous teeth (SHED) (18) after approval by the Brazilian National Research Ethics Committee (Protocol number CAAE 12892419.0.0000.5347). SHED characterizations were performed by morphological analysis, flow cytometry, and differentiation assay *in vitro* (18–20).

To analyze cell viability on the scaffolds, the membranes were placed into a 24 well plate. SHED and HaCaT cells were seeded at a density of  $25 \times 10^3$  cells per well and incubated at 37 °C with 5 % CO<sub>2</sub>. The culture media DMEM (low and high glucose for SHED and HaCaT, respectively) were supplemented with 10 % fetal bovine serum and 1 % penicillin/streptomycin. After 1 and 6 days, the cells were treated with 0.25 µg/ml of MTT for 4 hours at 37 °C. The supernatant was removed and 200 µl DMSO was added per well to dissolve the formed formazan crystals. The last procedure was made twice to obtain a good dissolution of the crystals. Absorbance was measured at 560 nm and 630 nm with the equipment Multiskan™ FC Microplate Photometer (Thermo Scientific™).

Cytotoxicity of the PLGA/Fib scaffolds was also analyzed by indirect MTT. The scaffolds were placed in a sterile tube and 1 ml of the culture medium with 1 % antibiotic without fetal bovine serum was added to each sterile sample. The scaffolds were incubated and after 7 days, the

scaffold incubated medium (SIM) was removed and supplemented with FBS 10 %. DMEM high glucose supplemented with FBS 10 % was considered as the control medium (CM). The cells (1600 HaCat per well) were seeded and cultivated in a 96-well plate and incubated for 8 hours to allow for the cells to adhere to the plate. After the incubation period, the culture medium was removed and 200  $\mu$ l of SIM or CM was added to each well.

After 24 hours, the cell culture medium was removed and 50  $\mu$ l of MTT with a concentration of 0.5 g/ml was added to each well. The plate was placed in a 5 % CO<sub>2</sub> incubator at 37 °C for 2 hours. The media was then discarded and 200  $\mu$ l DMSO was added to each well to dissolve the formazan crystals. The absorbance was measured at 570 nm. The toxicity and cell viability of the scaffold was calculated as follows:

$$\text{Toxicity \%} = \left( 1 - \frac{\text{mean of } A_{SIM}}{\text{mean of } A_{CM}} \right) \times 100 \quad \text{Viability \%} = 100 - \text{Toxicity\%}$$

Where  $A_{SIM}$  is the absorbance of the cells treated with SIM and  $A_{CM}$  is the absorbance of the cells treated with the control medium.

### 2.5.8 Cell adhesion and morphology

The presence of cells attached to the nanostructured scaffolds was confirmed by fluorescence microscopy. The scaffolds were analyzed on the 1st and 6th days after the keratinocytes and SHED were seeded on the membranes. The membranes were washed in PBS, fixed in 4 % paraformaldehyde buffer for 30 minutes, rinsed in buffer PBS and the cell nuclei were blue stained with 0.5  $\mu$ g/ml of DAPI (1 minute). To evaluate the morphology, the cells were stained with phalloidin (30  $\mu$ g/ml) for actin filaments. The photographs were obtained by fluorescence microscope Leica DMI8 (Leica Microsystems).

### 3. Results and discussion

#### 3.1 Morphology and diameter of the nanofibrous scaffolds

Fig. 2 shows by SEM the formation of randomly oriented nanofibers with heterogeneous diameters. The SEM images of the PLGA scaffolds are illustrated in Figs. 2a, b; the mean diameter of the fibers was  $1051.0 \pm 290.2$  nm (Fig. e).

The PLGA/Fib scaffolds presented a diameter ranging from 200 nm to 1300 nm, with a mean of  $639.8 \pm 241.8$  nm. Approximately 77 % of the fiber diameter was located between 300 and 800 nm (Figs. 2c, d, f). The smaller diameter in the PLGA/Fib scaffolds is probably related to the addition of  $\text{CaCl}_2$  salt to the plasma, improving the electrical conductivity of the polymeric solution and, consequently, decreasing the fibers diameter (21).

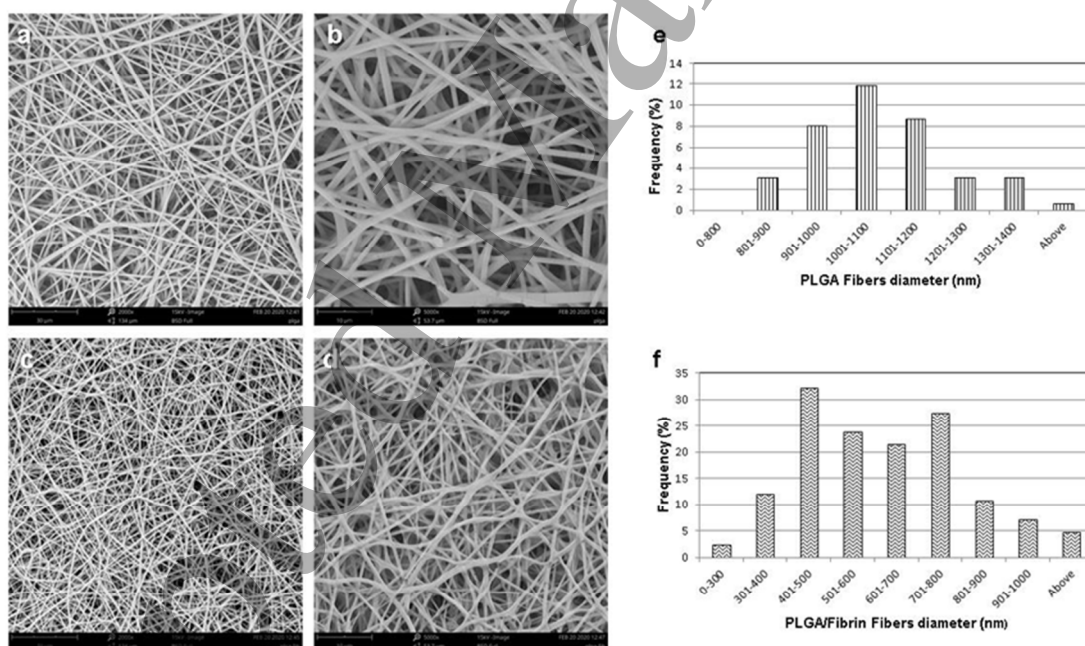


Fig. 2. SEM characterization for nanofibrous scaffolds. PLGA scaffold (a, b). PLGA/fibrin scaffold (c, d). Diameter distribution for PLGA scaffolds (e). Diameter distribution for PLGA scaffolds (f).

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3 It is important to highlight that fibers with concentrations of PLGA lower than 40 % presented  
4 beads and it was necessary to increase their concentration for obtaining uniform fibers and to  
5 counteract the acidic hydrolytic effect of formic acid. Formic acid could be responsible for  
6 maintaining a higher percentage of fibers with a diameter in the nanoscale (22). On the other  
7 hand, it was observed that by increasing the concentration of the natural polymer, the fibers also  
8 presented beads; for this reason it was necessary to maintain a 1 % fibrin concentration. Formic  
9 acid has been used to improve the solubilization of natural polymer and to develop hybrid  
10 scaffolds; the presented results are consistent with the findings of other investigations when  
11 formic acid was used. Shalumon and colleagues developed a hybrid scaffold of chitosan and  
12 PCL using formic acid as a solvent, and to avoid the formation of beads, it was necessary to  
13 increase the concentration of the synthetic polymer and decrease the concentration of the natural  
14 one (23). However, studies using low concentrations of the natural polymer have already  
15 reported efficient results (24)(25)(26)(27), and in this study,, it is demonstrated that the  
16 incorporation of 1 % fibrin increases cell viability.  
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### 36 **3.2 Thermal stability and water contact angle**

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38 The hydrophilicity of the nanofibrous membranes was analyzed by contact angle determination.  
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40 The water contact angle of the PLGA scaffolds was  $118.9 \pm 2.9$  (Fig. 3a); however, with the  
41 incorporation of fibrin, the water contact angle of the PLGA/Fib scaffolds decreased to  $111.1 \pm$   
42  $2.8$  (Fig. 3b). Although fibrin presents a moderated hydrophobic nature after fibrinogen  
43 polymerization, which is determined by acid-base and Van der Waals interactions, the water  
44 contact angle of the PLGA/fibrin scaffolds was reduced (28). While it can still be considered  
45 hydrophobic, a lower water contact angle could influence the improvement of cell-scaffold  
46 interaction.  
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55 To investigate the thermal stability of the PLGA/Fib scaffolds, TGA analysis was carried out.  
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57 On the TGA curves (Fig. 3c) it is possible to see a shift of the curve to the left, which signifies  
58 that the incorporation of fibrin facilitates scaffold degradation.  
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3 The PLGA scaffold showed two decomposition events, the first between 30-100 °C, probably  
4 related to residual solvent evaporation (Table 1, Fig. 3d). The second event occurred between  
5 100-500 °C when 98.3 % of the scaffold weight was lost; this is associated with thermal  
6 degradation of the polymer backbone, as seen in other investigations (29)(30).  
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11 The PLGA/fibrin scaffold showed four degradation events. The first event, related to the  
12 residual solvent, appears again between 30-100 °C, and 89.3 % of the scaffold weight had been  
13 lost between 100-380 °C. The third degradation event for the PLGA/Fib scaffold occurred  
14 between 380-450 °C, when 5.6 % of the weight was lost. This event is probably related to fibrin  
15 degradation.  
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22 The fibrin substrate presented slower degradation kinetics, with four degradation events. In this  
23 case, weight loss occurred gradually, without big peaks of degradation. The polymer had lost  
24 62.8 % of its weight at 550 °C, and above this temperature, the final decomposition phase or  
25 carbonization of the polymer was observed (31). A residual product of 28.9 % at 800 °C was  
26 seen (Table 1, Fig. 3d). The result shows that isolated fibrin is a stable product with important  
27 intermolecular forces, and explains why in the PLGA/Fib scaffold there was a residual mass at  
28 800 °C of 0.1 %, which is related to the natural polymer and its final carbonization phase (Table  
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40 Furthermore, based on the experience of our research group working with PLGA 75:25, it has  
41 been observed that the molecular weight of electrospun PLGA scaffolds decreased about 32%  
42 after 45 days of incubation in PBS (32) and an *in vivo* partial degradation of the PLGA scaffold  
43 after 6 weeks was also observed, (33). On the other hand, biodegradation of fibrin is mediated  
44 by plasmin and it is subject to a variety of factors such as tissue metabolism rate, tissue depth,  
45 presence of cells and fibrinogen concentration (34). Wolbank and colleagues reported that the  
46 subcutaneous degradation kinetic for fibrin was of approximately 16 days (35). Similarly, other  
47 authors have reported fibrin degradation in between one and two weeks, depending on the  
48 addition or not of cells (36) (37) (38). It has therefore been suggested that, due to the  
49 degradation characteristics of PLGA spun scaffolds, they should be used for chronic wounds  
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(39). However, future studies should be made to determine the influence on the scaffold degradation behavior when 1% of fibrin is incorporated on PLGA mats.

Table 1. Characteristic degradation temperatures of PLGA/fibrin scaffolds and fibrin substrate

Sample	Events	Temperature Range (°C)	T max (°C)	Weight Loss (%)	Weight (%)				
					100 °C	300 °C	500 °C	700 °C	Residue at 800 °C
PLGA	1	30-100	72.3	0.32	99.68	96.08	1.38	0.02	0.0
	2	100-500	386.1	98.29					
	3	500-800	-----	1.39					
PLGA/Fib	1	30-100	72.1	2.31	97.69	85.63	2.26	0.26	0.1
	2	100-380	353.4	89.31					
	3	380-450	402.8	5.56					
	4	450-800	-----	2.72					
Fibrin	1	30-200	158.8	12.60	96.92	76.40	39.02	32.52	28.94
	2	200-360	320.6	30.88					
	3	360-550	375.4	19.28					
	4	550-800	-----	8.30					

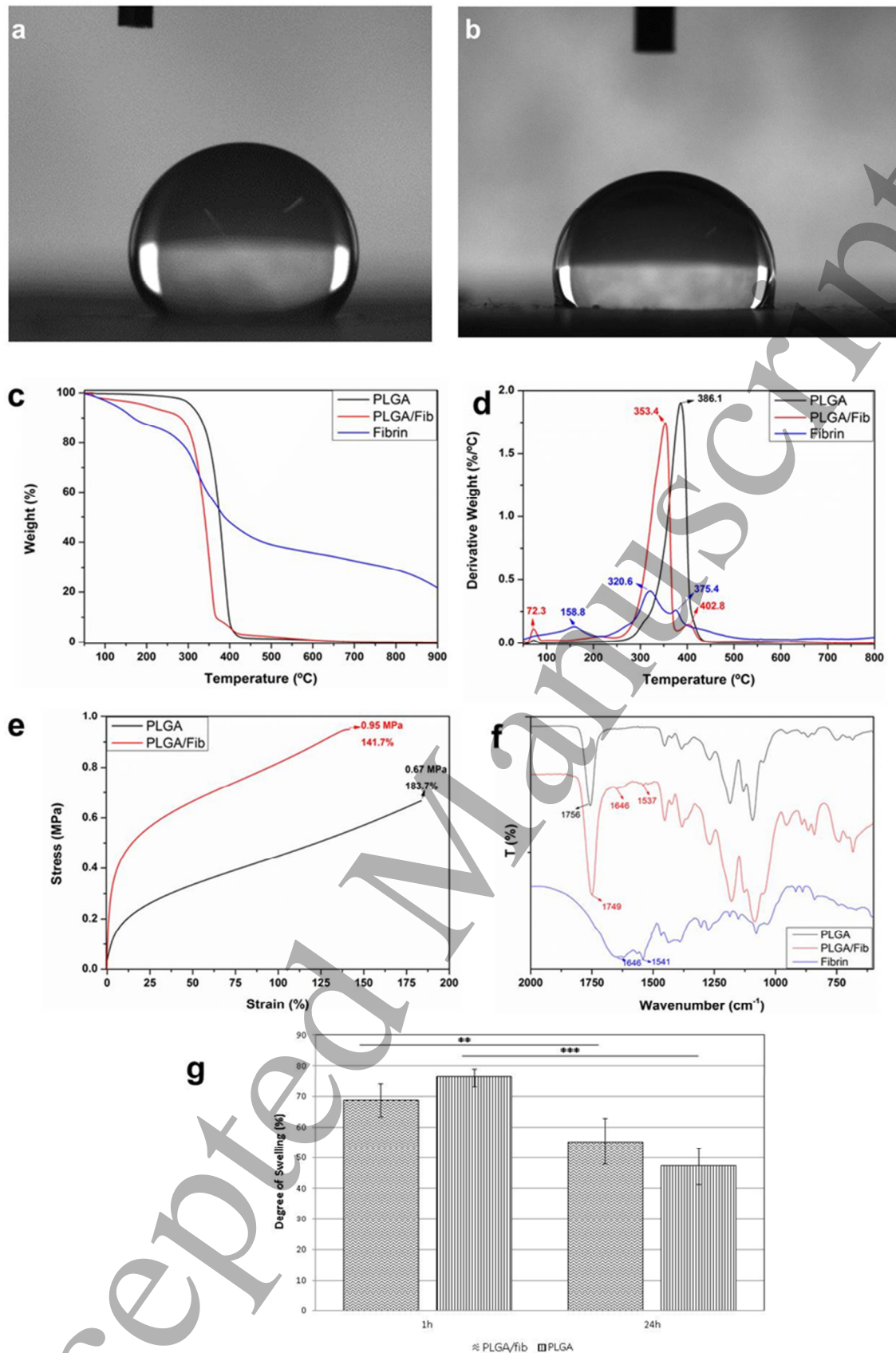


Fig. 3. Water contact angle for PLGA scaffold (a) and PLGA/fibrin scaffold (b). TGA thermograms (c). Derivative of the TGA curve (d). Stress-strain curves (e). Fourier transform infrared spectroscopy (f). Degree of swelling for scaffolds (g).



### 3.3 Mechanical properties of nanofiber scaffolds

Stress-strain curves of PLGA and PLGA/fib scaffolds are presented in Fig. 3e. The tensile strength of the scaffold increased when fibrin was incorporated, meanwhile, elongation at break decreased from 183.7 to 141.7 %. The increment of the tensile strength and decrease of elongation at break could be attributed to the fibrin's intramolecular hydrogen bonding properties, inducing bundle formation by reinforcing the PLGA fibers (24). Studies have found values of higher tensile strength and lower elongation at break for PLGA scaffolds, even with a lower concentration of the polymer (40)(41)(29), which indicates that the effect of formic acid on the polymer should be further studied.

Thus, it can be concluded that fibrin incorporation enhances the mechanical properties of the scaffold and such properties are expected to improve with the incorporation of cells and the gradual deposition of the extracellular matrix.

### 3.4 Fourier transform infrared spectroscopy of scaffolds

Influences of fibrin in the PLGA scaffolds were analyzed in the range between 2000 and 750  $\text{cm}^{-1}$ . FTIR results are shown in Fig. 3f. The fibrin presented characteristic amide absorption peaks (N-H), including 1541  $\text{cm}^{-1}$  (Amide II bond) and 1646  $\text{cm}^{-1}$  (Amide I bond) (42)(43). PLGA displayed a characteristic absorption carbonyl (C=O) pick in 1756  $\text{cm}^{-1}$  as well as a characteristic absorption band between 1100–1250, which represents the esters groups (C-O) (44). On the other hand, the PLGA/fibrin scaffold showed similar peaks of both fibrin and PLGA.

Although the association of fibrin with synthetic polymers using the electrospinning technique for the development of scaffolds has already been reported, these studies used commercial fibrinogen (45)(46) or purified fibrinogen (47), which incurs higher production costs. Some studies have supported the use of blood plasma, but in these cases, the intention has been of coating the electrospinning mats with fibrin (48)(49), and not including the natural polymer in the electrospinning solution.

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3 The method used in this present study, therefore, is advantageous because of its simplicity and  
4 low-cost, it being possible to construct the mat with autologous plasma without the risk of  
5 rejection induced by the natural polymer.  
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### 10 11 12 13 **3.5 Swelling behavior**

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15 The ability to absorb water is a relevant characteristic of a skin substitute because of the  
16 importance of preventing severe fluid loss in full-thickness injuries. This study calculated the  
17 swelling degree of the mats, to evaluate if the incorporation of fibrin could contribute to  
18 decreasing the swelling degree of the scaffold, taking into account that fibrin from a plasma  
19 blood clot is moderately hydrophobic (28). The results showed no statistically significant  
20 difference at the first hour for the PLGA and PLGA/fibrin scaffold, with a swelling degree of  $76$   
21  $\pm 3 \%$  and  $69 \pm 5 \%$ , respectively Fig. 3g. However, after 24 hours, both mats presented a  
22 reduction in the swelling capacity of  $55 \pm 7 \%$  for the PLGA/fibrin and  $47 \pm 6 \%$  for the PLGA.  
23 Although the PLGA/fibrin swelling degree was better, the difference was not significant. The  
24 behavior of the membranes was expected due to the phenomenon of contraction that PLGA  
25 spun fibers undergo when mats are incubated in PBS at  $37 \text{ }^\circ\text{C}$ , a phenomenon that has been  
26 widely reported (50)(51). The contraction forces of the polymer fibers could thus hinder the  
27 scaffold expansion caused by water absorption.  
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### 46 47 48 **3.6 Blood compatibility**

49 To evaluate if the PLGA/fibrin scaffold was compatible with blood, the hemolysis percentage  
50 was calculated. The hemolysis percentage denotes the ratio of erythrocytes, whose cell  
51 membrane is broken down when it comes into contact with the mat. Thus, a lower hemolysis  
52 percentage suggests good hemocompatibility of the scaffold. Fig. 4a illustrates the absorbance  
53 of the hemolysis assay. Absorbance for the positive control was  $0.432 \pm 0.012$ . Absorbance for  
54 the negative control, PLGA scaffold and PLGA/fibrin scaffolds was:  $0.006 \pm 0.001$ ;  $0.006 \pm$   
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0.001, and  $0.006 \pm 0.001$ , respectively, with hemolysis percentage for the PLGA and PLGA/fibrin scaffolds of 0.117 and 0.088, respectively. It can be concluded that both scaffolds were highly hemocompatible.

### 3.7 Indirect MTT Assay

To test the cytocompatibility and toxicity of the mats, an indirect MTT assay was performed. Fig. 4b shows increased cell viability of  $107.2 \pm 6.8$  % in the cells cultivated with the SIM of PLGA/fibrin mats compared with the control (cells cultivated with DMEM supplemented with FBS 10 %). The results suggest that the SIM of the PLGA/fibrin scaffold contributed to improving the viability of the HaCaT cells, perhaps due to the fibrin release from the scaffold. On the other hand, the toxicity percentage of the PLGA/fibrin scaffold was: -7.2 %. A negative result was obtained due to the mean of the absorbance in the group of cells cultivated with SIM from the PLGA/fibrin scaffold being greater than that of the control.

### 3.8 Direct MTT Assay

MTT assay was used to evaluate the cell viability of the HaCaT and SHED cells on the scaffolds after 1 and 6 days. The tissue culture plate (TCP) was taken as control. Figs. 4c and d illustrate absorbances for each group at different times.

The results showed similar viability of the HaCaT cells on the PLGA/fibrin scaffolds compared with the TCP on the first day (Fig. 4c). In contrast, the PLGA scaffold promoted significantly reduced viability compared with the PLGA/fibrin scaffolds. On the sixth day, there was a significant difference in the HaCaT cell viability cultivated on the PLGA/fibrin scaffold compared with the PLGA scaffold, although, it was inferior to the TCP. The results suggest that the incorporation of fibrin influenced initial HaCaT adhesion and proliferation. Although it was described that keratinocytes are not able to bind to fibrin due to the lack of integrin  $\alpha v \beta 3$  (52), it has also been described that migration and proliferation of keratinocytes is dependent on fibrinogen concentration (53); fibrinogen concentration, therefore, ranging from 17–33 mg/ml in

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3 fibrin 3D constructs provides an appropriate proliferation for keratinocytes. On the other hand,  
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5 in the protein composition of a fibrin clot, fibronectin has been identified (43) (54), which has  
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7 adhesion sites for  $\alpha 5\beta 1$  integrins, expressed in keratinocytes, being important for keratinocyte  
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9 migration (55). Besides this, it has been suggested that fibrin stimulates keratinocytes to  
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11 secrete TGF- $\alpha$ , which in turn increases cell proliferation and EGF receptor capacity, supporting  
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13 the proliferation of keratinocytes in an autocrine fashion (56).  
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17 When SHED viability was evaluated, the results showed a significant difference between all the  
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19 groups, both on day 1 and day 6 (Fig. 4d). On the first day, there was a significantly increased  
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21 viability of SHEDs on the PLGA/fibrin scaffold compared to the PLGA scaffold; however, the  
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23 viability in the TCP was superior. In contrast, on day 6, the PLGA/fibrin scaffold had a  
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25 significantly increased viability compared to the TCP as well as the PLGA scaffold. The PLGA  
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27 scaffold viability was significantly reduced when compared to the PLGA/fibrin scaffold and the  
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29 TCP.  
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33 The results suggest that the incorporation of fibrin in the mats improved the biological  
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35 properties of the PLGA polymer, and viability on the mats is cell type-dependent.  
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### 40 **3.9 Cell adhesion and morphology**

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42 The nuclei of the cells were stained with DAPI to evaluate if the cells were able to adhere to the  
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44 scaffolds after one day and, to evaluate the morphology, actin filaments were stained with  
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46 phalloidin (Figs. 4e-j). The SHED cultivated on the PLGA/fibrin scaffolds (Fig. 4g) presented a  
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48 similar conformation of their morphology comparable with the control group (Fig. 4e).  
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50 Meanwhile, the SHED culture in the PLGA scaffolds stills exhibited rounded morphology (Fig.  
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52 4f). The HaCaT cells presented a similar cytoskeleton conformation in both the PLGA and  
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54 PLGA/Fib scaffolds (Figs. 4i and j, respectively), being evident the presence of a higher cell  
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56 density on the scaffold with fibrin incorporation. Images were taken in sites of greater cellular  
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58 confluence.  
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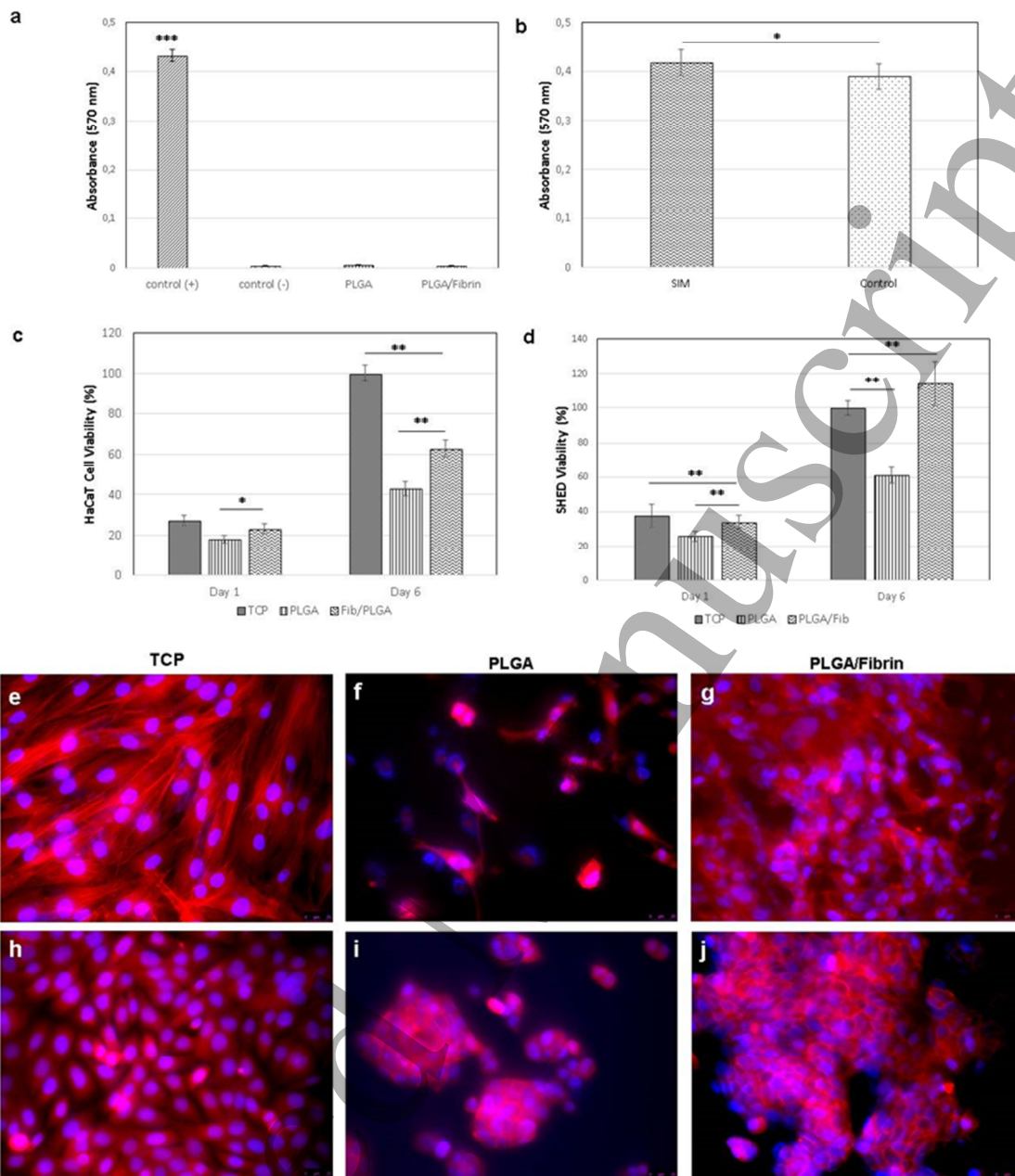
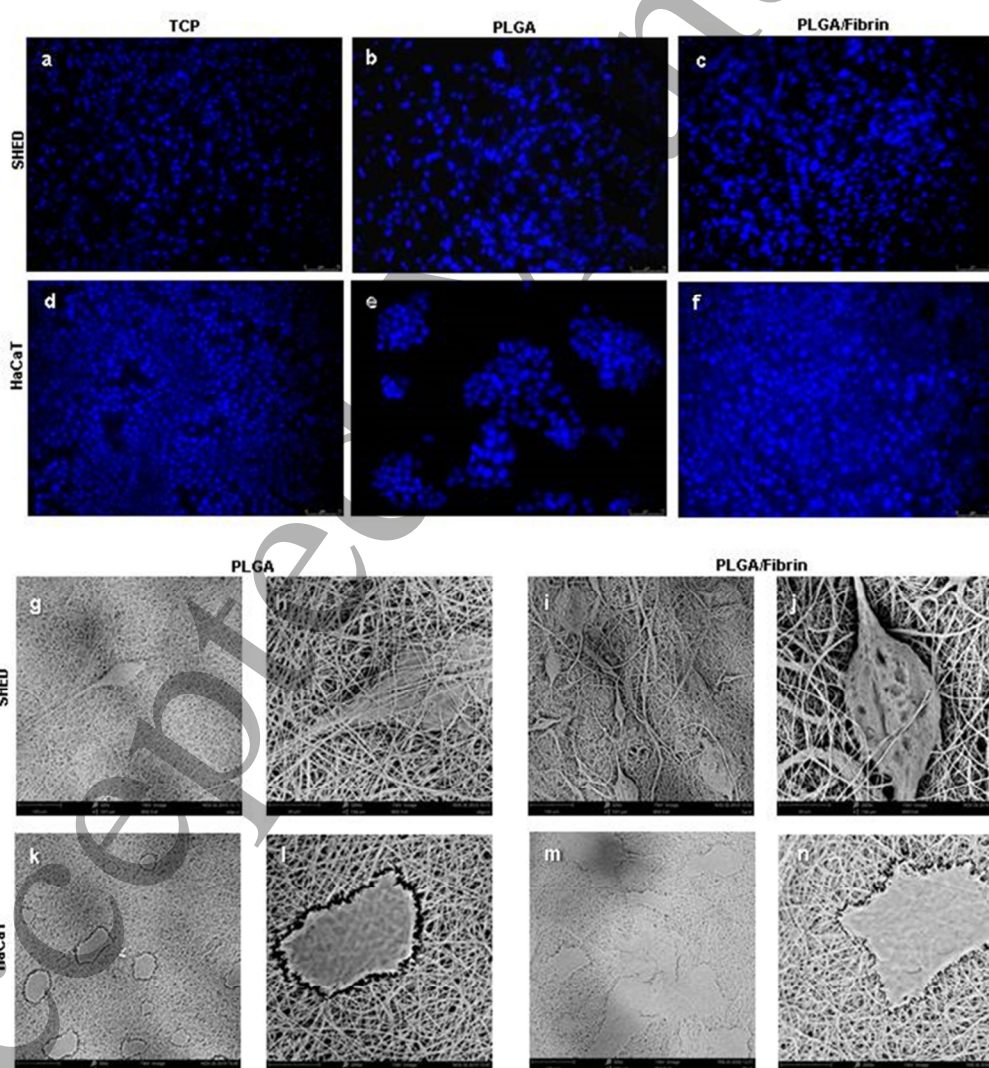


Fig. 4. Hemolysis assay for scaffolds (a). Indirect MTT assay (b). MTT assay of scaffolds using HaCaT cells at days 1 and day 6 (c). MTT assay of scaffolds using SHED cells at days 1 and day 6 (d). DAPI and phalloidin stain on scaffolds at day 1, with 400x magnification using SHED (e, f, g) and HaCaT cells (h, i, j).

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3 The images taken at day six with DAPI stain are shown in Figs. 5a-f. Magnification 200x shows  
4 an overview of cell density on day 6, which confirms the results shown in the MTT assay. A  
5 higher nuclei density for both cell types in the PLGA/fibrin scaffolds can be seen compared  
6 with the PLGA scaffolds (Figs. 5c and 5f).  
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12 As seen in the DAPI stain images, the SEM images of 500x magnification (Figs. 5g, i, k, and m)  
13 showed a greater cell number in both cell types cultivated in the scaffolds with fibrin  
14 incorporation. It is even possible to see a layer of HaCaT cells on the PLGA/fibrin scaffold (Fig  
15 5m), meanwhile, scattered cells are seen in the PLGA scaffolds (Fig. 5k). The images with  
16 2000x magnification (Figs. 5h, j, l, n) show a normal morphology of both cell types at day six,  
17 on both the PLGA and PLGA/fibrin scaffolds.  
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3 Fig. 5. DAPI staining cell nuclei on scaffolds at day 6, with 200x magnification using  
4 SHED (a, b, c) and HaCaT cells (d, e, f). SEM images for scaffolds cultivated with cells at  
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7 day 6, using SHED (g, h, i, j) and HaCaT cells (k, l, m, n).  
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#### 10 11 12 4. Conclusion

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14 In the present investigation, nanofibrous hybrid scaffolds of PLGA synthetic polymer and  
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16 fibrin natural polymer were produced. It was demonstrated that the incorporation of 1 %  
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18 fibrin allows for the production of material with enhanced mechanical properties and cell  
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20 viability as well as good blood compatibility. Furthermore, the isolation method of fibrin  
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22 was shown to be a simple and efficient method for the obtainment of fibrin and probably  
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24 other plasma proteins, which improve the biological properties of PLGA scaffolds. It can  
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26 therefore be suggested that fibrin could be used for spinning other synthetic polymers,  
27  
28 considering the poor biological properties of such materials. Finally, it can be concluded  
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30 that the PLGA/fibrin scaffold is a promising material to be used as a skin substitute.  
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